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وجلة علهية نصف سنوية وحكوة

تصدر عن كلية التربية - جسامعـة ذمـــار



- إدمان الألعاب الإلكترونية وعلاقته بالشعور بالمسؤولية وتقدير الذات والتواصل الأسري لدى طلبة المرحلة الثانوية بمدينة نجران
 - الأثار القرآنية الإيمانية والأمنية والطبية والنفسية والأخلاقية والاجتماعية
 دراسة موضوعية
 - الثبات على الحق في سورة آل عمران- دراسة تفسيرية موضوعية
- الجَوِهَرة الوَفِيَّة، والدَّرة السَّنِّية في الكَلام، في إيضاَح ما نَقَلَهُ الخَفاجِي من عبارة إبن الهُمِام، تأليف: محمد بن يوسف جدي (المتوفى:1345 هـ) ضبط نصفها، وقدم لها، وحققها الباحثان: عادل معيلي، ومرتضى مصنوم
- الضوابط والتنبيهات على الأخطاء الشائعة في التلاوة عند المقرئ جمال الدين الملحاني (ت938هـ)

المجلة العلمية لكلية التربية

مجلة علمية نصف سنوية

تعنى بالدراسات والبحوث الإنسانية والعلمية المختلفة ـ تصدر عن كلية التربية ـ جامعة ذمار

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المجلة العلمية لكلية التربية

تعنى بالدراسات والبحوث الإنسانية والعلمية المختلفة تصدر عن كلية التربية جامعة ذمار الجمهورية اليمنية

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مجلة علمية نصف سنوية ـ تصدر عن كلية التربية ـ جامعة ذمار ـ الجمهورية اليمنية ، محتوياتها متاحة مجانا لكل الباحثين والقراء ، وتسمح للجميع بالطباعة والتنزيل والتوزيع ومشاركة النص للمقال كاملا دون اجتزاء ، واستعمالها في الأغراض العلمية والبحثية بالإشارة إلى مؤلفيها.

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- يكتب البحث بلغة سليمة، ويراعى فيه قواعد الضبط ودقة الأشكال -إن وجدت- في صيغة Word ويكتب البحث بخط Sakkal Majalla وحجم 15 بالنسبة إلى الأبحاث باللغة العربية، وهوامش بحجم 11، وخط Sakkal Majalla فحجم 14، وهوامش بحجم 12، وتكون العناوين الرئيسة بخط غامق، و حجم 14، على أن تكون المسافة بين الأسطر 1 سم، ومسافة الهوامش 2,5سم من كل جانب.
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- يوضح الباحث هدف البحث، والمنهجية، وأهم نتيجة في الملخص (على ألا يزيد الملخصان بالعربية والإنجليزية، كل منهما عن 170 كلمة، ولا يقل عن 120 كلمة، في فقرة واحدة، ويرفق معهما كلمات مفتاحية بحيث لا تزيد عن 6 كلمات.



- المقدمة: يحتوي البحث على مقدمة يستعرض فيها الباحث: نبذة عن الموضوع، الدراسات السابقة، ثم الجديد الذي سيضيفه البحث في مجاله، إشكالية البحث، أهدافه، أهميته، ومنهجه، وخطة سيره في بحثه، بشكل مترابط ومتسلسل.
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- يتم ترتيب المصادر والمراجع ألفبائيًا، على أن لا يدخل في الترتيب أل، وأبو، وابن، فابن منظور مثلا يرتب في حرف الميم.
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 - التحقّق النهائيّ، ويُمنح الباحث خطاب قبول بالنشر، متضمنا رقم العدد الذي سوف ينشر فيه وتاريخه.

- -بعد التأكد من جاهزية المخطوطة بصورتها النهائية، يتم إرسالها إلى التدقيق اللغوي والمراجعة الفنية، ثم تحال إلى الإنتاج النهائي.
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Structural, Optical, and Electrical Properties of Pure α-Al₂O₃ Nanopowders with Add V₂O₇ and Cu₂O by Sol-Gel Method

Cu2O التركيبية والضوئية والكهربائية لمساحيق α -Al2O3 النانوية النقية مع إضافة ν 2O7 و ν 2O9 الخواص التركيبية والضوئية والكهربائية السوجل

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ملخص البحث:

هدف هذا البحث إلى بيان التوليف والتوصيف الشامل لـ α-Al2O3)8-x(V2O7)20(Cu2O)x النقى و α-Al2O3)8-x(V2O7)20(Cu2O)x مولية متفاوتة%1 = x) ، %2، و3%) في شكل مسحوق نانوي تم تحضيره بطريقة السول جل. وبعد إخضاع العينات للمعالجة الحرارية، اكتشفنا تحييد الأشعة السينية (XRD) عن درجة تبلورها العالية. وتم حساب الحجم البلوري باستخدام صيغة شيرل، وأظهرت اختلافات بين العينات ذات α-Al2O3 النقى الذي يمتلك حجم 43.10 نانومتر ومساحيق الأكسيد النانوبة المختلطة التي تقع في نطاق 29.54 نانومتر إلى 34.84 نانومتر. وأظهرت هذه المواد خصائص أشباه الموصلات، إذ تتراوح قيم فجوة النطاق من 3.28 فولت إلى 5.21 فولت. علاوة على ذلك، مع زبادة التركيز المولى للنحاس، تحسنت الموصلية الكهربائية، مما يدل على التطبيقات المحتملة في الأنظمة التي تتطلب سلوكًا كهربائيًا منظمًا. وأهم نتائج البحث تتمثل في تسلط الضوء على تنوع هذه المواد، مما يوفر فرصًا لبلورة مخصصة، وضبط فجوة النطاق، والخواص الكهربائية الخاضعة للتحكم، ليجعلها مناسبة للتطبيقات التكنولوجية والعلمية المتنوعة.

الكلمات المفتاحية: الخصائص الهيكلية والبصرية والكهربائية α-Al2O3/VO/Cu)؛ مساحيق النانو.

Abstract

This manuscript explored the synthesis and comprehensive characterization of pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)_{8-x(}V₂O₇)₂₀(Cu₂O)_x at varying molar ratios (x = 1%, 2%, and 3%) in nanopowder form, prepared using the solgel method. After subjecting the samples to heat treatment, X-ray diffraction (XRD) revealed their high crystallinity. crystallite size was calculated using the Scherrer formula. exhibited among the samples with pure α-Al₂O₃ possessing a size of 43.10 nm and the mixed oxide nanopowders falling within the range of 29.54 nm to 34.84 nm. These materials displayed semiconductor characteristics, with band gap values ranging from 3.28 eV to 5.21 eV. Moreover, as the molar concentration of copper increased, electrical conductivity improved, signifying potential applications in systems electrical behavior. requiring regulated These findings highlight the versatility of these materials, offering opportunities for tailored crystallinity, band gap tuning, and controlled electrical properties, making them suitable for diverse technological scientific applications.

 $\begin{tabular}{ll} \textit{Keywords} : & Structural, & Optical, & and \\ Electrical & Properties; & α-Al_2O_3/VO/CuO; \\ nanopowders & \end{tabular}$



1. Introduction

Metal oxide nanoparticles have gained widespread prominence due to their remarkable properties and their myriad applications in both optical and electrical fields [1]. Among these metal oxides, aluminum oxide (Al₂O₃) stands out as a versatile ceramic material extensively employed in various applications, including catalysts, transparent armor for ballistic instruments, lasers, discharge lamps, and infrared (IR) airborne sensors [2]. Notably, Al_2O_3 exists in multiple structural forms, denoted by symbols like α , κ , θ , γ , β , χ , δ , and $\dot{\eta}$ [3]. Among these, α -alumina oxide (α -Al₂O₃ or corundum) holds a distinguished place due to its thermodynamic stability and finds applications in transparent armor for ballistic protection, ceramics, highstrength materials, adsorbents, catalysts, catalyst supports. performance field-effect transistors (FETs), electrical insulators, optoelectronics, thermoluminescent dosimeters, lasers, light-emitting displays, cutting tools, spark plugs, and gas sensing [2,4-6].

 $\alpha\text{-Al}_2\text{O}_3$ forms when exposed to temperatures above 1100 °C, adopting a hexagonal crystalline structure with lattice parameters a=4.758 and c=12.991 [1,7]. This phase exhibits a direct energy transition and energy gap (Eg) ranging from 4.116 to 8.8 eV [4,8]. Despite its excellent thermal properties, $\alpha\text{-Al}_2\text{O}_3$ is less favorable in terms of optical and electrical characteristics. However, a previous study [1] has shown that blending $\alpha\text{-Al}_2\text{O}_3$ with V2O5 and CuO can enhance its structural properties while reducing the energy gap (Eg) and improving its electrical properties. Mixed oxide nanopowders containing $\alpha\text{-Al}_2\text{O}_3$, V2O5, and CuO have found applications in diverse fields.

Among the vanadium oxide phases, V_2O_5 is the most stable and possesses a high oxidation state. Its distinctive structural properties make it valuable for applications such as solar cells, gas sensors, optical-electrical switches, chemical sensing, and electrochromic optoelectronic devices [9]. V2O5 has a direct energy gap (Eg) ranging from 2.2 to 2.8 eV, and it crystallizes in orthorhombic and tetragonal structures with lattice parameters a=3.561, b=11.501, and c=4.378 [4,10].

Copper oxide (CuO) is a p-type semiconductor with a relatively small energy gap (Eg) ranging from 1.2 to 1.9 eV. It adopts a monoclinic

crystalline structure with lattice parameters a=4.69, b=3.42, and c=5.13 [11]. Research in this area has predominantly focused on nanomaterials with enhanced structural, optical, and electrical properties, synthesized through various methods such as sol-gel [4,1], hydrothermal [12], co-laser vaporization [3], and physical vapor deposition (PVD) [13].

In this study, we employ the sol-gel method to synthesize pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)_{8-x}(V₂O₇)₂(Cu₂O)_x with x values of 1%, 2%, and 3%. Our investigation delves into the crystallinity size, optical properties, and electrical characteristics of these α -aluminum oxide nanopowders with different molar ratios. The aim is to harness the exceptional properties of aluminum oxide while incorporating V₂O₅ and CuO to enhance its optical and electrical attributes. Ultimately, this research explores the potential for tailored aluminum oxide materials that can find applications in various fields.

2. Experimental details

2.1. Materials

In this study, various materials were employed, including aluminum nitrate (Al (NO₃)₃·9H₂O) (HIMEDIA, 95%), copper nitrate rehydrate (Cu (NO₃)₂·3H₂O) (HIMEDIA, 99%), ethanol (C₂H₅OH) (SEGMA, 96%), and ammonium monodentate (NH₄VO₃) (HIMEDIA, 95%).

2.2. Synthesis method

The nanopowders of pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)_{80-x}(V₂O₇)₂₀(Cu₂O)_x with x values of 1%, 2%, and 3% were synthesized using the sol-gel method, as outlined below.

2.2.1. Synthesis pure α-Al₂O₃

To prepare pure α -Al₂O₃, 15g of Al (NO₃)₃·9H₂O was dissolved in 40 ml of ethanol to achieve a 1M solution at room temperature. The solution was stirred using a magnetic stirrer for 20 minutes until it became homogeneous. The temperature was then increased to 80 °C while maintaining continuous stirring, resulting in the formation of a gel solution. This solution was left in the beaker for 24 hours and subsequently dried in an oven at 180 °C for 2 hours. The resulting material was ground until it reached a soft powder consistency.



2.2.2. Synthesis $(\alpha-Al_2O_3)_{80-x}(V_2O_7)_{20}(Cu_2O)_x$

For the synthesis of $(\alpha-Al_2O_3)_{80-x}(V_2O_7)_{20}(Cu_2O)_x$, 10.50g of Al (NO3)3·9H2O was dissolved in 40 ml of ethanol to create a 0.7M solution at room temperature. Simultaneously, 935.8mg of ammonium monovanadate (NH₄VO3) was dissolved in 40 ml of ethanol to produce a 0.2M solution, and 966.4mg of $Cu(NO_3)_2$ ·3H2O was dissolved in 40 ml of ethanol to achieve a 0.1M solution. All these solutions were stirred with a magnetic stirrer for 20 minutes until they reached homogeneity. The homogeneous solutions were mixed together while stirring for 20 minutes at room temperature until a uniform solution formed. This mixture was then stirred for an additional 20 minutes at 80 °C to produce a gel, which was left in a beaker for 24 hours. Afterward, the gel was dried in an oven at 180 °C for 2 hours.

All samples underwent calcination at 1200 °C for 5 hours, and pellets for electrical measurements were prepared. The pellets were made using a pressing machine (Carver) under a pressure of 6000 kg, resulting in pellets with a diameter (d) of 13 mm and a thickness (L) of 2 mm.

2.3. Characterizations

The structural properties of the samples were assessed through X-ray diffraction (XRD) using a Shimadzu model 6100-EDX-720 with CuK α radiation at 1.54065 Å. XRD patterns were obtained in the angular range of 0–80°. The optical properties of the samples were examined using a UV-Vis spectrophotometer, specifically the Hitachi U3900 model.

3. Results and discussion

3.1. Structure Properties

In the analysis of the structural properties, we began with the examination of pure α -Al2O3, which exhibited characteristic X-ray diffraction (XRD) patterns, as illustrated in Figure 1. A series of distinct diffraction peaks of α -Al₂O₃ were identified at 20 angles of 25.50°, 35.20°, 37.70°, 43.30°, 52.60°, 57.46°, 61.30°, 66.50°, and 68.20°. These peaks corresponded to the atomic planes of (012), (104), (110), (113), (024), (116), (018), (214), and (300), respectively. The XRD pattern for pure α -Al2O3 closely matched the hexagonal crystal structure with space group R-

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3c, consistent with standard JCPDS cards, No. 00-46-1212 and PDF #10-0173

In the second sample, where the molar concentration of aluminum was reduced to 80% by the introduction of 20% vanadium, the intensity of XRD peaks for α -Al₂O₃ diminished, as depicted in Figure 1. This decrease in α -Al₂O₃ intensity can be attributed to the inclusion of impurity material V₂O₅, potentially leading to crystal disorder and deformation within pure α -Al₂O₃. Notably, new peaks emerged, indicating the presence of V₂O₅. These new peaks corresponded to standard reflection patterns in JCPDS card No. 00-45-1074 and PDF #41-1426, and they were observed at 20 angles of 12.10°, 26.20°, 27.80°, and 28.80°, aligning with the (200), (330), (240), and (241) planes of V₂O₅, respectively.

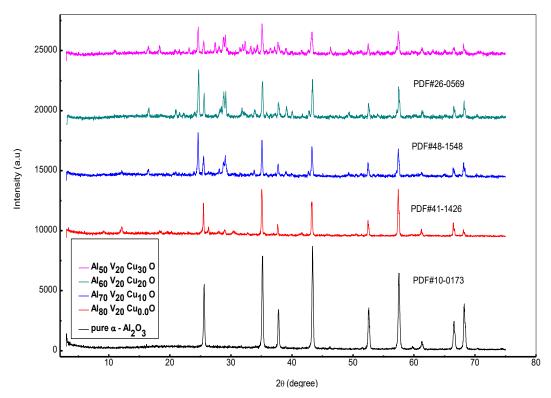


Fig 1. XRD pattern a high crystallization nanopowders of pure α -Al₂O₃ and (α -Al₂O₃)_{80-x} (V_2O_7)₂₀(Cu_2O)_x at x = (1,2 and 3 %).

In the third sample, the molar concentration of aluminum was further reduced to 70%, while introducing a 10% molar concentration of copper



while keeping the vanadium concentration consistent. The XRD patterns displayed a similar trend, with diminished α -Al2O3 intensity attributed to the introduction of CuO as an impurity, resulting in weak crystallinity. The CuO peaks were not distinct due to their weak crystalline nature. The 2 θ values of 12.1°, 20.3°, 26.2°, and 39.9° aligned with (001), (110), (111), and (311) planes, respectively, and corresponded to JCPDS card No. 00-41-1426 and JCPDS card No. 00-26-0569. Furthermore, at 2 θ values of 16.30°, 20.70°, 21.50°, 24.45°, 27.30°, 32.0°, 35.70°, 43.26°, and 49.30°, new peaks were observed, corresponding to (110), (111), (-112), (200), (112), (-133), (-221), (-131), and (133) crystalline planes of β -Cu₂V₂O₇. This formation was a result of thermal solid–solid interactions between copper and vanadium oxide phases in the mixed oxide nanopowders [2].

In the fourth sample, the molar concentration of copper was increased while decreasing the molar concentration of aluminum to 60%. This alteration led to an increase in the intensity of β -Cu₂V₂O₇ peaks, indicating improved crystallinity. Additionally, a new peak of CuO was observed at 20 values of 38.90°, corresponding to (200) planes. This finding was consistent with the standard JCPDS card No. 00-48-1548. Furthermore, the CuO peak appeared at 20 values of 61.30°, corresponding to the (-113) plane, and aligned with JCPDS card No. 00-045-0937 and PDF#26-0569, confirming its presence as per prior research [3,4]. In the fifth sample, with an increased molar concentration of copper up to 30%, new peaks were observed, indicating a further change in the crystalline structure.

The crystallite size of $(Al_2O_3)_{80-x}(V_2O_7)_{20}$ (Cu₂O)_x at x = (1, 2, and 3%) was calculated using the Scherrer equation from Full Width Half Maximum (FWHM), as outlined in Equation.

$$D = \frac{0.89 \,\lambda}{\beta \, \text{COS}\theta} \tag{1}$$

Where λ represents the wavelength of the X-ray, θ indicates Bragg's angle, and β shows the FWHM. The results are summarized in Table 1. It was observed that the crystallite size gradually decreased as the molar concentration of copper increased, transitioning from (43.10 nm) for pure α -Al2O3 to (29.54 nm) for (Al2O3)80-x (V2O7)20 (Cu2O) x at x = (1, 2, and 3%) in mixed oxide nanopowders, as presented in Figure 2. This decrease can be attributed to the fact that the ionic radius of aluminum oxide (0.54 Å)

is smaller than that of copper oxide (0.67 Å) and vanadium oxide (0.78 Å). Interestingly, the crystallite size of α -Al2O3 also decreased as the molar concentration of copper increased, indicating that the secondary phase's evolution influenced the particle size of the parent phase (α -Al2O3) during crystallization. These results align well with prior research [5].

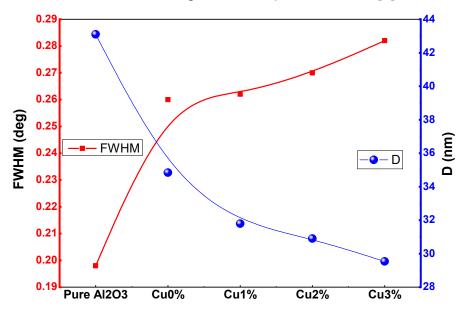


Fig 2 Variation of full-width-at-half-maximum and crystallite sizes.

Table 1 pure α -Al₂O₃ and (Al₂O₃)_{80-x} (V₂O₇)₂₀(Cu₂O)_x at x = (1, 2 and 3 %) Nanopowders with were synthesized by sol-gel method.

	Phase	2θ		FWH	d (Å)		£	
Samples	oxide	(deg)	hkl	M (deg)	Expt	Std	a(Å)	c(Å)
Pure α- Al ₂ O ₃	α-ΑΙ2Ο3	43.30	113	0.198	2.085	2.085	4.756	12.99 3
Al ₈₀ V ₂₀ Cu _{0.0} O	α-Al ₂ O ₃	57.00	116	0.260	1.601	1.601	4.117	13.02 1
Al ₇₀ V ₂₀ Cu ₁₀ O	α-Al ₂ O ₃	35.00	104	0.262	2.561	2.550	4.115	13.00
20 3 4 10 5	β-	24.46	200	0.528	3.636	3.600	7.330	10.20



	Cu2V2O7							0
Al ₆₀	α-Al ₂ O ₃	35.08	104	0.270	2.556	2.550	4.111	13.00 3
V ₂₀ Cu ₂₀ O	β- Cu2V2O7	24.54	200	0.272	3.625	3.600	7.249	09.64 0
Al ₅₀	α-Al ₂ O ₃	35.04	104	0.282	2.558	2.552	4.110	13.03
V ₂₀ Cu ₃₀ O	β- Cu2V2O7	43.20	-133	0.321	2.092	2.088	7.220	09.70 0

Furthermore, the decrease in crystallite size led to an increase in the dislocation density (δ), calculated using Equation (2) [6].

$$\delta = \frac{1}{D^2} \tag{2}$$

As summarized in Table 2, These results suggest that defects and vacancies became more prominent with the increasing molar concentration of copper.

Additionally, we examined the lattice constants (a, b, and c) for the dominant phases, α -Al₂O₃ and β -Cu₂V₂O₇, using equations (3) and (4) for hexagonal structures. The unit cell volume (V) and strain (ϵ) were calculated using equations (5) and (6), with the calculated values provided in Table 2. The unit cell volume and strain for all samples and the lattice constants were in good agreement with others [7]. Notably, the lattice constant (a) decreased as the molar concentration of copper increased, a trend that can be attributed to the diminishing values of ionic radius and the consequent decrease in crystallite size.

$$\frac{1}{d^2} = \frac{h^2}{a^2} + \frac{k^2}{b^2} + \frac{l^2}{c^2} \tag{3}$$

$$\frac{1}{d^2} = \frac{1}{\sin^2 \beta} \left(\frac{h^2}{a^2} + \frac{k^2 \sin^2 \beta}{b^2} + \frac{l^2}{c^2} - \frac{2hl \cos \beta}{ac} \right) \tag{4}$$

$$V = 0.866 ca^2 \tag{5}$$

$$\epsilon = \frac{\beta \cos \theta}{4} \tag{6}$$

This comprehensive analysis of structural properties sheds light on the intricate interplay of materials in these mixed oxide nanopowders and their impact on crystallinity and lattice characteristics.

Table 2 Structure properties of the α -Al₂O₃ / V₂O₅/CuO at different molar ratio of Al⁺³ and Cu⁺¹

Complex	Phase oxide	ala	V	D	δ*10^15	Strain
Samples Phase oxid		c/a	$(\mathring{\mathbf{A}}^3)$	(nm)	$(m)^{-2}$	(3)
Pure α-Al ₂ O ₃	α-Al ₂ O ₃	2.73	254.51	43.10	5.38	0.046
Al ₈₀ V ₂₀ Cu _{0.0} O	α-Al ₂ O ₃	3.16	191.13	34.84	8.24	0.057
Al ₇₀ V ₂₀ Cu ₁₀ O	α-Al ₂ O ₃	3.16	190.66	31.79	0.99	0.063
70 20 - 10 -	β-Cu ₂ V ₂ O ₇	1.39	474.60	29.23	1.17	0.129
Al ₆₀ V ₂₀ Cu ₂₀ O	α-Al ₂ O ₃	3.16	190.31	30.90	1.05	0.064
00 20 - 20 -	β -Cu ₂ V ₂ O ₇	1.33	438.68	29.88	1.12	0.066
Al ₅₀ V ₂₀ Cu ₃₀ O	α-Al ₂ O ₃	3.17	190.64	29.54	1.15	0.067
20 20 20 30 2	β-Cu ₂ V ₂ O ₇	1.34	437.89	26.60	1.41	0.075

3.2. Optical properties

The optical properties of pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)_{80-x} (V₂O₇)₂(Cu₂O)_x at $x=(1\%,\ 2\%,\ and\ 3\%)$ nanopowders, each with varying molar ratios, were meticulously examined. The transmittance spectra of these samples were measured across a wavelength range spanning from 200 nm to 800 nm. Notably, the transmittance levels were particularly high in the vicinity of 300 nm. However, beyond this wavelength, the transmittance values began to decline, as depicted in Figure 3. For instance, the (Al₈₀V₂₀Cu_{0.0}) sample displayed a transmittance of approximately 90%. This relatively high transmittance could be attributed to the specific coloration characteristics of this particular sample. In contrast, as the molar concentration of copper increased in the (Al₇₀V₂₀Cu₁0) sample, the transmittance soared to approximately 95%, indicating improved light transmission. Continuing this trend, further increases in the molar concentration of copper in both the (Al₆₀V₂₀Cu₂₀) and (Al₅₀V₂₀Cu₃₀) samples resulted in transmittance levels of around 60% and 45%, respectively. This suggests a clear correlation

between the presence of copper and the optical properties of the mixed oxide nanopowders, with increased copper concentrations leading to enhanced transmittance.

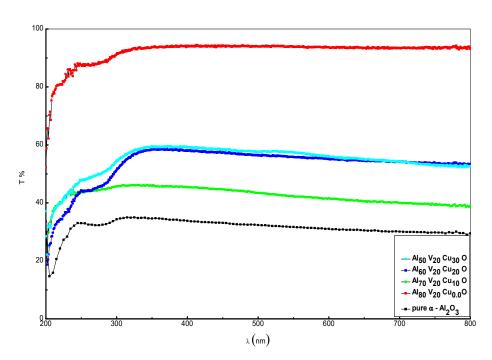


Fig 3 UV spectra for transmittances with wavelengths (λ nm) for all samples.

The determination of band energy in semiconductor materials is often facilitated through optical absorption studies. Figure 4 showcases the absorption characteristics of pure α -Al₂O₃, where it becomes evident that absorption intensifies with the increase in wavelength. Notably, the absorption edge of pure α -Al₂O₃ is pinpointed at approximately 280 nm, a finding that aligns with prior research [8]. Extending the investigation to the mixed oxide nanopowders, we observe the absorption spectra shown in Figure 4. In these spectra, distinct absorption edges become apparent for each sample, specifically at 331 nm, 344 nm, 364 nm, and 387.5 nm. These values correspond to the samples Al₈₀V₂₀Cu_{0.0}O, Al₇₀V₂₀Cu₁₀O, Al₆₀V₂₀Cu₂₀O, and Al₅₀V₂₀Cu₃₀O, respectively. These absorption edges are significant as they signify the photoexcitation of electrons from the valence band to the conduction band within the samples. Furthermore, the absorption bands are attributed to electronic transitions occurring from the

occupied 2p bands of oxygen to the unoccupied 3d bands of vanadium [9]. This optical absorption study offers valuable insights into the band energy characteristics of the mixed oxide nanopowders, shedding light on their semiconductor properties and potential applications.

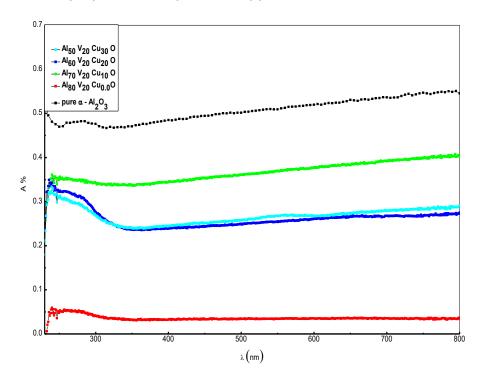


Fig 4 The absorption of pure and with different concentrations.

To provide deeper insights into the experimental data, we employed theoretical Equation (8) [16], to fit the observed results

$$\alpha h \nu = B(h \nu - E_g)^{\frac{1}{n}} \tag{8}$$

Equation 8, which relates the absorption coefficient (α) to various parameters, such as Planck's constant (h), incident light frequency (v), a characteristic parameter (B), and the band gap energy (Eg), also incorporates the variable exponent (n). The quality of the fit was found to be optimal when n was set to 2, suggesting that the transitions observed are directly allowed.

The band gap energy (Eg) was determined for all the samples, including pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)_{80-x}(V₂O₇)₂₀(Cu₂O)_x at x = (1%, 2%, and



3%). The obtained band gap values were as follows: 5.21 eV for pure α-Al₂O₃, and 3.62 eV, 3.44 eV, 3.38 eV, and 3.28 eV for the respective mixed oxide nanopowders. Notably, the band gap value of 5.21 eV for pure α-Al₂O₃ closely aligns with prior research, confirming the reliability of the measurement, as shown in Figure 5.

Furthermore, an interesting trend emerged when examining the impact of varying the concentration of aluminum oxide and copper within the nanopowders. With a reduction in the concentration of Al2O3 and a simultaneous increase in the molar concentration of copper, the band gap of the nanopowders decreased. This trend is visually depicted in Figure 6 and quantitatively summarized in Table 3. The decrease in the band gap energy holds significance for potential applications, as it implies changes in the electronic properties of these materials, further expanding their utility in various technological and scientific domains.

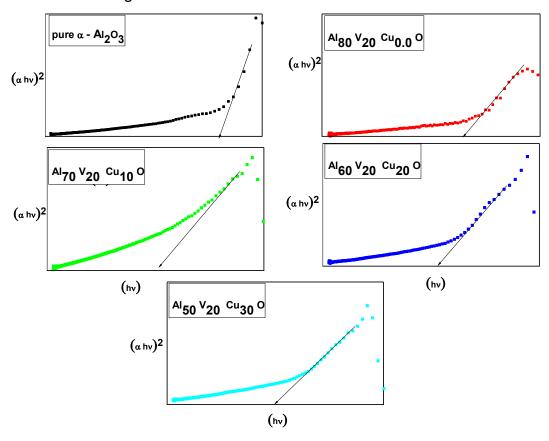


Fig 5 Plots of the $(\alpha hv)^2$ versus the photon energy (hv) of the pure and with Cu % nanopowders for all samples.

The reduction in the band gap value observed can be attributed to the emergence of empty levels induced by defects situated within the band gap. This phenomenon is well-documented, where the band gap of any material is intricately controlled by the concentration of defects it harbors. In the case of α -Al2O3, two types of defects play pivotal roles: donor defects, represented by oxygen vacancies, and acceptor defects, which are denoted as Al interstitials. These defects introduce energy levels situated below the conduction band and above the valence band, respectively. The genesis of these energy levels can be traced back to two fundamental reactions. Firstly, the Al interstitial defects are the result of the Frenkel reaction, a process where an atom is displaced from its original lattice site and occupies an interstitial position within the crystal structure. On the other hand, oxygen vacancy defects emerge from the Schottky reaction, which involves the simultaneous creation of vacancies for both cations and anions within the crystal lattice [10].

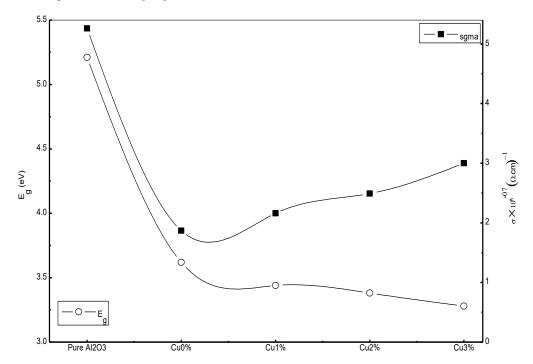




Fig 6 Variation of both the energy gap (E_g) and electrical conductivity(σ)at α-Al₂O₃ and Cu (1,2, and 3 %).

The presence and concentration of these defects and their intricate interplay fundamentally influence the band gap of the material. This understanding of defect-induced alterations in the band gap offers valuable insights into the electronic properties of the mixed oxide nanopowders, providing a foundation for the exploration of their potential applications in various scientific and technological contexts.

3.3. Electrical properties

The electrical properties of the materials were further assessed by measuring the current-voltage (I-V) characteristics and determining the ohmic resistance (R) for both pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)_{80-x}(V₂O₇)₂₀(Cu₂)_x at x=(1%, 2%, and 3%) nanopowders, each with distinct molar ratios. The results unveiled a clear trend in the ohmic resistance, where it consistently decreased with the progressive increase in the molar concentration of copper, as detailed in Table 3. This decrease in ohmic resistance signifies a corresponding increase in electrical conductivity (σ), a trend visually presented in Figure 6. The σ can be expressed as a function of the R, based on the $\sigma = L/A \times (1/R)$, with units in $(\Omega \text{ cm})^{\Lambda}$ -1. The variation in electrical conductivity is governed by the changing ohmic resistance, where a decrease in resistance leads to enhanced conductivity. A pivotal aspect to consider is that the σ within the sample is influenced by multiple factors. One of these factors is the movement of free electrons, a phenomenon closely associated with the presence of copper within the material [11]. The introduction of copper facilitates and eases the movement of free electrons, consequently resulting in higher electrical conductivity [12].

This observation highlights the profound influence of the molar concentration of copper on the electrical properties of the mixed oxide nanopowders. It signifies the potential for tailored electrical characteristics in these materials, which can be harnessed for specific applications requiring controlled conductivity and electrical behavior.

Table 3 Optical energy gap and electrical conductivity of pure α -Al₂O₃ and mixed oxide.

Samples	E _g (eV)	R (kΩ)	σ (Ω.cm) ⁻¹
Pure α-Al ₂ O ₃	5.21	28.6 E ⁹	5.26 E ⁻¹²
Al ₈₀ V ₂₀ Cu _{0.0} O	3.62	769.3	1.87 E ⁻⁰⁷
Al ₇₀ V ₂₀ Cu ₁₀ O	3.44	666.7	2.16 E ⁻⁰⁷
Al ₆₀ V ₂₀ Cu ₂₀ O	3.38	579.8	2.49 E ⁻⁰⁷
Al ₅₀ V ₂₀ Cu ₃₀ O	3.28	479.2	3.00 E ⁻⁰⁷

The obtained electrical conductivity (σ) values in this study fall within the conventional range typically associated with semiconductors, which typically spans from 10^4 to $10^9 \,\Omega^{-1}$.cm⁻¹ [13]. Notably, these findings are consistent with established values for α-Al₂O₃, which typically exhibits an average electrical conductivity of $(6.87 \times 10^{-12} \pm 1.22 \times 10^{-14} \Omega^{-1}.\text{cm}^{-1})$ [14]. This alignment underscores the reliability and validity of the measurements. Furthermore, the electrical conductivity values obtained for the mixed oxide nanopowders, particularly for CuO and V₂O₅, also bear similarity to welldocumented values in the literature. For instance, CuO exhibits electrical conductivity ranging between $(1.1\times10^{-4} \text{ and } 2.77\times10^{-4} \Omega^{-1}.\text{cm}^{-1})$ [15]. While V_2O_5 typically displays electrical conductivity in the range of $(2.53\times10^{-4}~\Omega^{-1})$ ¹.cm⁻¹) [16]. This consistency in the electrical conductivity values between the experimental findings and established values can be attributed, in part, to the ionic radii of aluminum (AI) and copper (Cu) present within the structural properties of the materials. The interaction and distribution of these elements within the material play a pivotal role in enhancing electrical conductivity [17]. These results provide valuable insights into the underlying mechanisms governing the electrical properties of the mixed oxide nanopowders, offering opportunities for applications that require controlled and predictable electrical behavior.



4. Conclusions

In this study, we successfully synthesized both pure α -Al₂O₃ and $(\alpha$ -Al₂O₃)₈₋ $_{x}(V_{2}O_{7})_{20}(Cu_{2}O)_{x}$ at x=(1%, 2%, and 3%) nanopowders, employing the sol-gel method, followed by heat treatment at 200 °C for 5 hours. The comprehensive examination of these materials yielded several noteworthy findings. XRD analyses demonstrated the high crystallinity of all samples, providing evidence of their welldefined structural characteristics. Notably, the crystallite size was accurately calculated using the Scherrer formula. For pure α-Al₂O₃, the crystallite size was determined to be 43.10 nm, consistent with its hexagonal structure. In contrast, for $(\alpha-Al_2O_3)_{80-x}(V_2O_7)_{20}(Cu_2O)_x$ in the mixed oxide nanopowders, the crystallite size exhibited variation, falling within the range of 34.84 nm to 29.54 nm. The band gap values for these materials also showed variation, ranging from 5.21 eV to 3.28 eV, indicative of the materials' semiconductor properties. Furthermore, the electrical conductivity exhibited a notable trend, with the resistance progressively decreasing as the molar concentration of copper increased. These findings collectively underline the potential and versatility of these mixed oxide nanopowders, particularly in applications that benefit from controlled crystallinity, .tunable band gap characteristics, and regulated electrical conductivity

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